

Performance of Ti₂AlC composite material in sports equipment

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Abstract

With the rising and development of material science, the application range of aluminum matrix composites is more and more extensive, and the requirements for performance are also higher. Ti₂AlC composites with both the properties of metals and ceramics have been extensively applied in various fields. In this study, highly purified Ti₂AlC which synthesized using atmospheric high-temperature calcination, and then it was mixed with Al powder and processed by pressureless sintering at 750 °C for 30 min; finally Ti₂AlC/Al composite was obtained. The obtained material was observed and analyzed using X-ray diffraction (XRD) and applied in the manufacturing of sports equipment. The surface abrasion was observed, and the frictional wear performance was analyzed. The results suggested that the friction coefficient and wear rate sharply decreased with the increase of the sliding speed, the emergence of surface oxide film and running-in; lamellar exfoliation and severe frictional wear appeared on the surface of the sports equipment when the sliding speed was low.

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Keywords: Ti₂AlC/Al composite; High-temperature sintering method; Pressureless sintering; Sports equipment; Friction performance

1. Introduction

With the development of society, ceramic production technology has been improved. Ti₂AlC has high strength, strong damage resistance, favorable electrical conductivity, oxidation resistance and low friction coefficient [1,2]. More and more ceramic materials are manufactured taking advantages of the mechanical and physicochemical properties of ceramics. Ti₂AlC as the material with the lowest density among ternary layered ceramic MAX-phase compounds has large heat conductivity coefficient and favorable electrical conductivity and antioxidant ability. Though Ti₂AlC is seldom reported in recent years, it has gradually raised the attention of experts. Yang et al. [3] revealed the concrescence of Ti₂AlC under the induction of oxidation and investigated the mechanism of oxidation driven concrescence and the factors affecting the concrescence efficiency of Ti₂AlC through experiments. Spencer et al. [4] investigated the reactivity of Ti₂AlC powders, with 3 and 10 μm of alumina, Al₂O₃, fibers in the process of pressure-assisted sintering. They manufactured samples using hot isostatic pressure (HIPed) or

hot pressing (HPed) and characterized them with differential thermal analysis, X-ray diffraction and electron microscope. It was found that Ti₂AlC could be reinforced by Al₂O₃, but the processing temperature should be kept below 1500 °C. After deep application of Ti₂AlC, it was found that the composites which are composed of Ti₂AlC and other metal materials has improved performance [5,6]. Zhu et al. [7] prepared Mo modified Ti₂AlC/AlO composite using reactive hot pressing. They found that (Ti, Mo)₂AlC/10wt%Al₂O₃ had a fine grain-size structure and that the Ickers hardness, bending strength, fracture toughness and compressive strength of the composite material were 4.75 GPa, 458 MPa, 6.03 MPa m^{1/2} and 971 MPa respectively. Yang et al. [8] successfully prepared TiAl/Ti₂AlC composites using in situ reaction hot-pressing taking Ti, Al and TiC powder as the raw materials, then investigated the organization and mechanical properties of TiAl/Ti₂AlC composites, and finally analyzed the phase change under different temperatures using X-ray diffraction (XRD). The results suggested that the density and hardness gradually increased with the increase of content of Ti₂AlC, and the fracture toughness and bending strength reached the peak values, 7.78 MPa m^{1/2} and 486 MPa, under the condition of 15wt%Ti₂AlC. In this study, Ti₂AlC/Al composite was developed using pressureless sintering at 750 °C and applied in

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the manufacturing of sports equipment. The friction and wear conditions of Ti_2AlC after being applied in sports equipment were evaluated through experiments.

2. Ti_2AlC

Ti_2AlC is a kind of ternary layered ceramic MAX-phase compound [9]. In Ti_2AlC crystal, Al atoms are separated from Ti octahedrons; the adhesion between Ti atom plane and Al atom plane is weak, which results in the self-lubricating characteristic of Ti_2AlC . MAX-phase composite has the properties of ceramics and metal, i.e., high melting point, high thermostability, favorable antioxidant activity and thermal shock resistance, low thermal expansion property, electrical conductivity, thermal conductivity, excellent machining property and favorable elasticity modulus and shear modulus [10]. Many sports equipment which is made of metals has oxidative exfoliation due to long-term exposure to air. Being applied in sports equipment, Ti_2AlC with strong oxidability can slow down oxidation on the surface of sports equipment and improve the utilization rate of sports equipment.

When sports equipment which is made from Ti_2AlC does high-speed movement, a large amount of oxide membranes will generate on the surface. Oxide membrane refers to a surface membrane generated when metals contact with components containing oxygen [11], and its main component is Fe. During friction, the oxide membranes on the surface of the material will have severe fracture and transfer under the effects of force and heat, resulting in exfoliation on the surface of the material. Oxide membrane has important impacts on the friction and wear properties of materials. When sports equipment is moving, good oxide membranes can prevent serious exfoliation on the surface of sports equipment and extend the service life of sports equipment.

3. Preparation and application of composite material

3.1. Preparation of Ti_2AlC under high temperature

There were two formulas. In formula 1, the molar ratio of Ti, Al, TiC and Sn was 1:1:0.9:0.1, and the percentages of Ti, TiC, Al and Sn were 34.05%, 19.19%, 38.32% and 8.44% respectively. In formula 2, the molar ratio of Ti, Al, TiC and Sn was 1:1:0.87:0.1, and the percentages of Ti, TiC, Al and Sn were 34.49%, 19.43%, 37.53% and 8.55% respectively.

The mass ratio of ratio of grinding media to material was 2:1. Agate ball was used as the grinding medium. The mass ratio of different ingredients are shown in Table 1. The weighed ingredients were put into a sealed plastic roller and processed by dry mixing using a ball tube mill at the speed of 100 r/min for 10 h.

Mixed powder was obtained after ball milling. Then it was put into a stainless steel mold and processed into round embryos with a diameter of 80 mm and thickness of 15 mm by cold pressing under 200 MPa. The embryos were transferred to a graphite mold which was smeared with BN powder and then roasted in a vacuum furnace. Before 200 °C, the vacuum

Table 1
The composition of the formulas.

No. of formula	1	2
Molar ratio of raw materials	Ti:Al:TiC:Sn = 1:1:0.9:0.1	Ti:Al:TiC:Sn = 1:1:0.87:0.1
Percentage of raw materials	Ti 34.05	34.49
	TiC 19.19	19.43
	Al 38.32	37.53
	Sn 8.44	8.55

furnace operated constantly to expel air, and the heating rate was kept at 30 °C/min. After 200 °C, argon was injected for protection. The sintering temperature was 1450 °C, and heat preservation lasted for 10 min.

Finally the furnace was turned down under the protection of argon.

Surface sticking substances were removed after the calcinations. Then the product was grinded to fragments in a diameter of 0.5 cm using a jaw crusher. Finally the fragments were processed by ball milling under the assistance of agate ball, powder and ethyl alcohol (2:1:0.7) using a high energy ball-milling machine at the speed of 200 r/min for 10 h.

3.2. Preparation of Ti_2AlC/Al composite

Ti_2AlC/Al composite was prepared by performing pressureless sintering after cold press molding of the powder which was mixed according as per formula 2.

Al powder and Ti_2AlC powder were mixed according to the ratio of 90 vs 10 vol%, 80 vs 20 vol%, 70 vs 30 vol% and 60 vs 40 vol% and then processed by dry mixing using a ball tube mill. After fully mixing, some of the mixture was transferred to a steel mold and pressed into embryos in a size of 15 mm × 15 mm × 20 mm at the pressure of 200 MPa. The formed embryos were put into a mold which was smeared with BN powder and then processed by vacuum sintering. Argon was injected for protection. The furnace temperature was increased to 750 °C at the heating rate of 30 °C/min. After heat preservation, the furnace temperature was decreased to 80 °C at the speed of 15 °C/min. Finally Ti_2AlC/Al composite was obtained. The XRD spectra of Ti_2AlC/Al composite are shown in Fig. 1.

It could be noted from Fig. 1 that the diffraction peak which was prepared at 750 °C had no remarkable difference no matter how the content of the two powders changed.

3.3. Experimental methods

As the composite would be applied in sports equipment, the possible wear modes of sports equipment were simulated using a SEARCHING friction testing machine. The friction mode was pin-on-disk sliding wear [12], and iron was taken for comparison. The sliding distance of the material was fixed, 20 m. The frictional wear test was performed by controlling the parameters such as sliding speed and load. The changes of frictional coefficient were collected using a computer which was connected to the equipment.

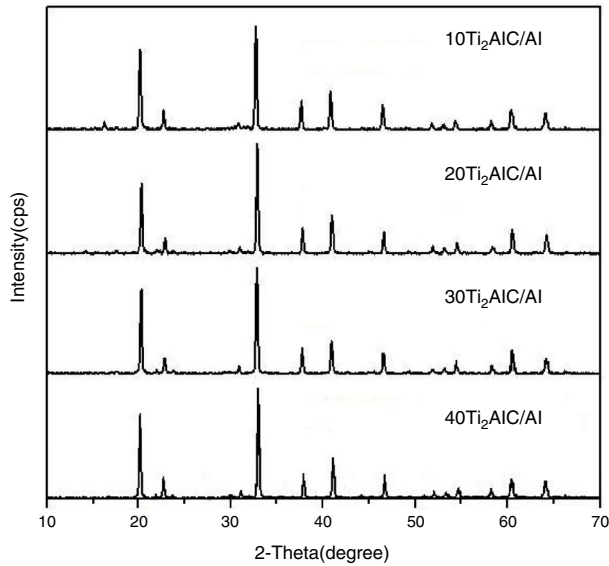


Fig. 1. The XRD spectra of $\text{Ti}_2\text{AlC}/\text{Al}$ composite.

Table 2

The friction coefficient of $\text{Ti}_2\text{AlC}/\text{Al}$ composite under different sliding speeds and loads.

Sliding speed	0.8 m/s	1.7 m/s	3.8 m/s	5.5 m/s
Friction coefficient when load was 20 N	0.79	0.76	0.68	0.42
Friction coefficient when load was 40 N	0.85	0.59	0.41	0.23
Friction coefficient when load was 60 N	0.59	0.42	0.38	0.22

The load was fixed at 40 N. The sliding speed of the material was set as 0.8 and 5.5 m/s. The frictional wear morphology was observed under a touch scanning electron microscope.

4. Experimental results and analysis

After the preparation of the experimental material, experimental analysis was performed. The experiment of frictional wear was performed through controlling parameters such as sliding speed and load.

The experimental material was tested when the load was 20, 40 and 60 N respectively. Moreover, the experimental data were recorded when the speed was 0.8, 1.7, 3.8 and 5.5 m/s.

As shown in Table 2, the friction coefficient gradually decreased as the sliding speed became higher; the friction coefficient decreased from 0.79 to 0.42 when the load was 20 N, from 0.85 to 0.23 when the load was 40 N, and from 0.59 to 0.22 when the load was 60 N, suggesting the same tendency under different load. Moreover it could be seen from the table that the material which bore lower load had larger friction coefficient; under the same sliding speed, the friction coefficient of the material decreased with the increase of the load.

As shown in Table 3, the wear rate of the material decreased firstly and then increased during the whole experiment; the wear rate was the lowest when the sliding speed was 5.5 m/s; the wear rate was 38×10^{-5} , 19×10^{-5} and $29 \times 10^{-5} \text{ mm}^3/\text{m}$

Table 3

The wear rate of $\text{Ti}_2\text{AlC}/\text{Al}$ composite under different sliding speeds and loads.

Sliding speed	0.8 m/s	1.7 m/s	3.8 m/s	5.5 m/s
Wear rate when load was 20 N ($10^{-5} \text{ mm}^3/\text{m}$)	38	10	5.2	5.1
Wear rate when load was 40 N ($10^{-5} \text{ mm}^3/\text{m}$)	19	4	3.8	0.8
Wear rate when load was 60 N ($10^{-5} \text{ mm}^3/\text{m}$)	29	12	8	5

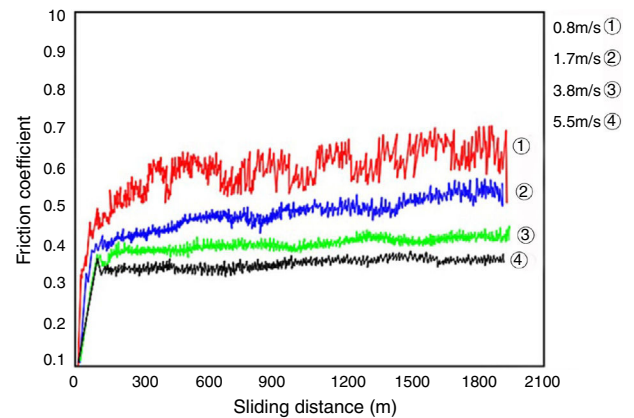


Fig. 2. The friction coefficient under different sliding distances and speeds.

respectively under 20, 40 and 60 N load respectively when the sliding speed was 0.8 m/s; when the load was 20 N, the wear rate of the material decreased from 38×10^{-5} to $5.1 \times 10^{-5} \text{ mm}^3/\text{m}$ with the increase of the sliding speed. It indicated that the sliding speed could severely affect the wear rate of the material.

The changes of the friction coefficient under different sliding speeds and distances were observed when the load was 40 N. It could be seen from Fig. 2 that the curve of the friction coefficient fluctuated dramatically when the sliding distance increased from 0.8 m/s; with the increase of the sliding speed, the fluctuation of the friction coefficient of the material became smaller gradually; when the sliding speed was 5.5 m/s, the fluctuation of the sliding coefficient of the material was the smallest. It suggested that the material started to running in at the beginning of the experiment; when the sliding speed was low, the running-in was not perfect enough and the friction coefficient increased constantly; with the increase of the sliding speed, the fluctuation of the friction coefficient became smaller, indicating a stable running-in stage.

In the study of the wear appearance of the external surface, a touch scanning electron microscope was used to observe the friction and adhesion appearance. The load was kept at 40 N, while the sliding speed was changed. The sliding speed was respectively set as 0.8 and 5.5 m/s. The scanning results are shown in Fig. 3. When the sliding speed was 0.8 m/s, there was distinct layer separation and exfoliation. When the sliding speed was 5.5 m/s, a white bright layer of oxide membrane formed on the surface of the material.

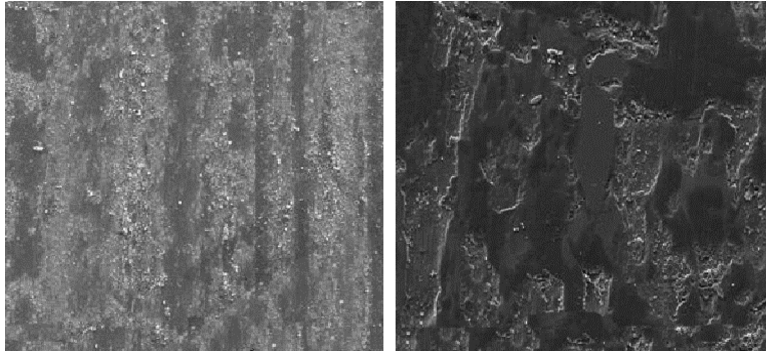


Fig. 3. Figures of wear appearance when the sliding speed was 0.8 m/s (left) and 5.5 m/s (right).

Ti₂AlC/Al composite has strong high-temperature resistance, high compressive strength and good lubricity. It is suitable to be applied in the manufacturing of sports equipment. Considering the frequent use of sports equipment in fitness exercises, its wear rate becomes the key for the manufacturing of such kind of equipment. Under the condition of dry friction, sliding speed and load are important factor affecting wear rate and friction coefficient. In this study, the frictional wear test was carried out when the load was 20, 40 and 60 N respectively and the speed was 0.8, 1.7, 3.8 and 5.5 m/s. The experiment suggested that the friction coefficient and wear rate decreased with the increase of the sliding speed; the wear rate was the lowest when the load was 40 N. The frictional wear morphology was observed under a touch scanning electron microscope. The surface of Ti₂AlC was oxidized when the sliding speed was 0.8 m/s, which made iron oxide leave on the surface; the major wear pattern was layered exfoliation. At the low speed, heat generated by friction resulted in oxidative wear on the surface of the material; however, there were few Fe oxidation films as the material had strong antioxidant activity and the temperature under low speed was low, which had a small impact on the wear rate. A low speed can cause strong adhesion, leading to layered exfoliation. When the sliding speed was 5.5 m/s, a white bright layer of oxide membrane formed on the surface of Ti₂AlC/Al composite. The oxide membrane with a loose structure and low hardness could lubricate Ti₂AlC. The oxide membranes on the surface of Ti₂AlC/Al composite distributed evenly when the sliding speed was 5.5 m/s, and Ti₂AlC/Al had a low friction coefficient and wear rate. It was assumed that the oxide membrane with lubrication effect effectively decreased the wear rate and friction coefficient. The emergence of the oxide membrane divided Ti₂AlC/Al composite and GCr15 and weakened the adhesion, which avoided exfoliation on the surface of the material. Moreover the lubrication effect of the oxide membrane significantly reduced risks of reoxidation after oxidative exfoliation and thereby reduced the wear rate. Ti₂AlC/Al composite is applied in the manufacturing of sports equipment such as rackets, bicycle and racing car because of its excellent friction property. Besides the cost of Al is lower compared to other metals such as Cu and Al₂O₃; hence the cost can be reduced when it is used for manufacturing sports equipment.

5. Conclusion

In this study, Ti₂AlC/Al composite was developed using pressureless sintering at 750 °C. Moreover the frictional wear test was carried out, and the wear conditions under different sliding speeds were observed. The results suggested that the friction coefficient and wear rate dramatically decreased with the increase of the sliding speed, the emergence of surface oxide film and running-in. When the sliding speed was slow, layered exfoliation and apparent friction and wear appeared. Therefore it can be used for manufacturing sports equipment used in high-speed sports. But the cost of the manufacturing of such sports equipment remains to be improved. Analyzing the cost effectiveness of production in a comprehensive aspect will be one of the research directions in the future.

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